Research Article

Pulse-Current Electrodeposition for Loading Active Material on Nickel Electrodes for Rechargeable Batteries

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Although the pulse-current electrodeposition method is a commonly used technique, it has not been widely employed in electrode preparation. This method was applied to sintered nickel electrodes in a nickel salt solution containing additives. The active material that was obtained, nickel hydroxide, was studied using different characterization techniques. Electrodes impregnated with pulse current had higher capacity than those impregnated with continuous current. The active material is homogeneous and compact with optimum loading and good performance during discharge. These characteristics would provide a large amount of energy in a short time due to an increase in the electrode kinetic reaction.

1. Introduction

Nickel hydroxide has many applications in the positive electrodes of alkaline cells such as nickel cadmium (Ni-Cd), nickel hydrogen (Ni-H₂), nickel metal hydride (Ni-MH), and nickel iron (Ni-Fe) cells [1–3].

The chemical pasting of a mixture of active material on a support conductor is used to prepare the positive electrodes of alkaline batteries [4]. This kind of electrode is used in the batteries of mobile phones, mp3s, emergency lights, and electric vehicles. Nickel hydroxide should have a homogeneous particle size and bulk distribution. Two methods are commonly used to obtain this kind of nickel hydroxide: indirect chemical precipitation and direct chemical precipitation. In both cases nickel hydroxide is obtained as a powder that then must be properly prepared to build the electrodes. For the manufacture of nickel hydroxide electrodes for use in high-performance batteries, such as nickel hydrogen batteries used as power source in space satellites, the impregnation technique used consists of the cathodic electrodeposition of sintered nickel plates. There are several variations of this technique such as (i) the Kandler process [5], which includes an electrochemical precipitation stage, working with a 0.3 M Ni(NO₃)₂ solution and an initial pH between 3 and 4, which is adjusted with nitric acid, and (ii) the process developed by Pickett and Maloy [6], which applies the same principle, but using a nickel nitrate solution in alcohol, pH = 3.5. These are the impregnation processes that were taken as the basis to develop our technique for the preparation of nickel hydroxide electrodes for high discharge capacity and long service life. The different properties of the electrodes that make them superior are not yet well known in the literature [7–9].

When the electrode is charged, nickel hydroxide is oxidized to nickel oxyhydroxide. During discharge the reaction is reversed, and Ni(OH)₂ has good cyclability and high energy density, both of which are important characteristics for alkaline batteries. To increase the energy density in each of these batteries, it is necessary to improve the performance of the positive electrode [10, 11]. The electrochemical energy storage in nickel hydroxide electrodes is related to the reversible characteristics of the redox couple nickel hydroxide/ox hydroxide, and in this work we describe how the electrode preparation method influences battery performance.

The Ni(OH)₂/NiOOH charge-discharge process is believed to be a solid-state, proton intercalation/deintercalation reaction [12], whereby both electrons and protons are exchanged and the processes are considered to be controlled by the bulk solid diffusion of protons [13]. If proton diffusion into the bulk solid phases was a slow process, Ni(OH)₂
buildup on the surface of the active material would cause a significant rise in polarization for the charge reaction. The accepted scheme for the Ni(OH)\(_2\)/NiOOH couple, which involves more than one phase, can be summarized as [14]

\[
\begin{align*}
\beta\text{-Ni(OH)}_2 & \xrightarrow{\text{Charge}} \beta\text{-NiOOH} + H^+ + e^- \\
\alpha\text{-Ni(OH)}_2 & \xrightarrow{\text{Dehydration}} \gamma\text{-NiOOH} + H^+ + e^- \\
\end{align*}
\] (1)

\(\beta\text{-Ni(OH)}_2\) has a well-defined crystalline structure and is the active material in chemically prepared nickel battery electrodes. In batteries, charging of the nickel hydroxide electrode involves the homogeneous solid-state oxidation of \(\beta\text{-Ni(OH)}_2\) to \(\beta\text{-NiOOH}\). Overcharging of \(\beta\text{-Ni(OH)}_2\), however, can lead to the formation of \(\gamma\text{-NiOOH}\).

The most widely used technique in the construction of electrodes for space battery applications is cathodic electrodeposition, where the precipitation occurs in one step [14–19]. The sintered nickel plate support is immersed in an aqueous solution of \(\text{Ni(NO}_3\text{)}_2\) (1.8 M) and a cathodic polarization current is applied. This produces an \(\alpha\text{-3Ni(OH)}_2\cdot2\text{H}_2\text{O}\) film [20], but \(\alpha\text{-Ni(OH)}_2\) is unstable in alkaline media and is easily converted to \(\beta\text{-Ni(OH)}_2\). When the positive current passes through the electrolytic cell, the solvent supporting the electrolyte system releases hydroxide anions. This may occur by the direct evolution of hydrogen gas at the electrode (2) or by reduction of the nitrate ion (3).

These anions can be produced on the negative electrode by the electrolytic decomposition of water:

\[
2\text{H}_2\text{O} + 2e^- \rightarrow H_2 + 2\text{OH}^- \quad E^0 = -0.83 \text{ V} \tag{2}
\]

or \(\text{OH}^-\) ions are produced by direct \(\text{NO}_3^-\) reduction:

\[
\text{H}_2\text{O} + \text{NO}_3^- + 2e^- \rightarrow \text{NO}_2^- + 2\text{OH}^- \quad E^0 = 0.01 \text{ V} \tag{3}
\]

An alternative reduction mechanism of nitrate to \(\text{NH}_4^+\) and \(\text{OH}^-\) ions has also been discussed in the literature [21, 22] (4):

\[
\text{NO}_3^- + 6\text{H}_2\text{O} + 8e^- \rightarrow \text{NH}_4^+ + 10\text{OH}^- \quad E^0 = -0.12 \text{ V} \tag{4}
\]

The reaction produces a hydroxide ion increase in the porous nickel plates, which then react with the nickel (5) present in the bulk solution to precipitate battery-active nickel hydroxide within the pores.

\[\text{Ni}^{2+} + 2\text{OH}^- \rightarrow \text{Ni(OH)}_2 \tag{5}\]

Metallic nickel may also be deposited within the pores of the plate material. In this way, its presence would presumably remove the active material from the electrode:

\[\text{Ni}^{+2} + 2e^- \rightarrow \text{Ni}^0 \tag{6}\]

During the immersion processes, \(\text{Ni(NO}_3\text{)}_2\) solutions with lower concentrations of metal nitrate are added in order to incorporate additives such as \(\text{Cd}, \text{Co}, \text{Zn}, \text{or Al}[23–29]\). Pure \(\text{Ni(OH)}_2\) is essentially an electrical insulator, so the effects of coprecipitated cobalt hydroxide increase the conductivity [30–32]. The cobalt ion addition to the active materials shows how \(\text{Co}^{+2}\) changes its oxidation state to \(\text{Co}^{+3}\) during the charge-discharge cycle. \(\text{Co}^{+3}\) cannot be reduced during subsequent discharging. There are also papers that discuss the addition of cobalt to nickel hydroxide electrodes, where an increase in the conductivity and the oxygen evolution potential and a delay in the mechanical failure of the electrodes have been reported [12, 29].

The aim of the present work was to determine whether the application of an intermittent train of pulses improves the electrochemical performance of nickel hydroxide electrodes. Only a few papers in the literature have explored these properties [33–35] and we want to know whether the structure of nickel hydroxide, which was deposited by the train of pulses, was responsible for giving nickel hydroxide a discharge capacity close to the expected theoretical maximum value.

2. Experimental

2.1. Electrochemical Impregnation of Nickel Hydroxide. The impregnation consists of an electrochemical process in aqueous solution. During this process many parameters must be controlled to keep the impregnation results consistent.

The temperature was controlled automatically at 70°C with the heating circulator. The solution contains 1.8 M nickel nitrate and 0.2 M cobalt nitrate.

The sintered nickel plate was pretreated in boiling distilled water for 15 min and then heated at 350°C for 30 min in order to remove impurities.

The sintered nickel plate (6.25 cm\(^2\)) was held in place by a Teflon fixture, maintained in a conventional glass cell in 6 M KOH solution that was prepared from AnalyR grade potassium hydroxide and twice-distilled water. The counterelectrode was a nickel mesh. A Hg/HgO reference electrode in 6 M KOH, which was the same as the measuring solution, was used as the reference electrode. Charge-discharge cycles at a 0.5 C rate (\(C = \text{theoretical capacity, 289 mAh}^{-1}\)) were performed (ARBIN BT 2000). All the potentials in this paper are referred to as the Hg/HgO electrode. The electrodes were discharged until 0.2 V; this potential was considered 100% of state of discharge (SOD).
### Table 1: Main characteristics of Ni(OH)₂ electrodes.

<table>
<thead>
<tr>
<th>Electrode/pulse (sec: sec)</th>
<th>Theoretical capacity (mAh)</th>
<th>Discharge capacity (mAh)</th>
<th>Specific capacity (mAh g⁻¹)</th>
<th>Utilization (%)</th>
<th>Loading (g cm⁻³)</th>
</tr>
</thead>
<tbody>
<tr>
<td>CP1 (on: 180/off: 90)</td>
<td>188.7</td>
<td>135.1</td>
<td>206.9</td>
<td>71.6</td>
<td>1.74</td>
</tr>
<tr>
<td>CP2 (on: 90/off: 45)</td>
<td>167.1</td>
<td>174.1</td>
<td>291.3</td>
<td>104.2</td>
<td>1.66</td>
</tr>
<tr>
<td>CP3 (on: 12/off: 6)</td>
<td>152.9</td>
<td>203.2</td>
<td>289.2</td>
<td>100.2</td>
<td>1.94</td>
</tr>
<tr>
<td>CP4 (on: 6/off: 3)</td>
<td>161.1</td>
<td>183.9</td>
<td>321.2</td>
<td>114.2</td>
<td>1.59</td>
</tr>
<tr>
<td>CC (continuous current)</td>
<td>197.4</td>
<td>157.1</td>
<td>229.9</td>
<td>79.5</td>
<td>1.9</td>
</tr>
</tbody>
</table>

(*) The theoretical capacity is calculated by considering the transfer of 1 electron (289 mAh g⁻¹) of active material.

Table 1 are at 70% charge/discharge cycles, 100% SOD.

3. Results

3.1. Galvanostatic Discharge Curve. After impregnation, the electrodes were rinsed in deionized water and formed using the Eagle-Picher formation procedure [37]. The procedure consisted of 5 1/2 cycles beginning and ending with the discharge. Each cycle was 40 min long in 20% potassium hydroxide. For the first 4 cycles, the current was at a 3.4 C rate. The current was reduced to 0.75 C for cycle five and again reduced to 0.5 C for the final half cycle. After that, the electrodes were cycled many times.

Figure 2 shows a contrast of the typical constant current discharge of the electrode impregnated with pulse current (CP3) and the electrode impregnated with continuous current (CC). Besides having greater capacity than CC (Table 1), the plateau for CP3 is better defined and is shifted by about 25 to 50 mV in the potential positive direction. This difference indicates that when the loading is high, the particles are not able to absorb the proton quickly; therefore, mass transport is the limitation. The higher the amount of active material is, the lower the active surface of the material results. Hence it is proton diffusion within the solid one that limits the speed of the reaction. For this reason, it has a greater ohmic drop.

3.2. Rate Capability. To study the performance of the electrodes, the response of the discharge capacity of the electrode versus different discharge currents was also taken into account. To estimate the electrode capacity, the discharge current values ranged from C/20 to 3 C in order to observe the behavior of the electrode under strict discharge conditions and lower current values.

We used different discharge currents, maintaining the charge current and the charging time, C/2, for 2.5 h.

Figure 3 shows an electrode that was impregnated with a pulse current (CP2 and CP3), delivering over 80% of the...
maximum capacity at a 2 C rate of discharge. This means that this type of electrode provides a large amount of energy in a few seconds. As a result, the utilization of active material was considerably improved and the electrode polarization resistance decreased after impregnating the electrode with pulse current.

Table 1 shows that the loading of all electrodes is similar but the utilization rate (discharge capacity/theoretical capacity) of the nickel electrode impregnated with continuous current is much lower than that of the other electrodes.

In some cases, a mass of active material was deposited but was not used in the discharge reactions.

For other electrodes, the number of exchanged electrons per metal atom (NEE) was greater than 1 (>289 mAh g⁻¹), a number that may be surprising. Since the reaction Ni(OH)₂/NiOOH involves the exchange of a unique electron, this observation comes from the fact that upon extended charging, 125% of the initial capacity assuming a number of NEE of 1, γ-NiOOH oxidized phase formation occurs and is characterized by an oxidation degree between 3.3 and 3.6 due to the presence of Ni⁴⁺ ions [38].

Audemer et al. [39] showed that the electrochemical capacity increases as the particle size decreases. A critical size seems to appear around 250 Å, which corresponds to the best electrochemically active hydroxide. When the particle size is small, the surface/volume ratio is high, so the amount of adsorbed water is larger.

The electrode impregnated with a high-frequency pulse current shows the best performance of all electrodes tested. This situation can be directly related to the morphology of the deposit of active material as evidenced in previous work [40].

A low current (C/20) of discharge produces self-discharge because there are parallel reactions [41].

3.3. Influence of Current Pulse on Discharge Capacity. The optimum electrode was chosen from different electrodes impregnated with different current frequencies. The cyclability and discharge capacity were taken into account as parameters by selection. These two parameters are very important in the performance of batteries. The electrode CP3 is one of the best electrodes and was selected to be compared with electrode CC.

So far, no relationship has been found between frequency and electrochemical performance.

3.4. Cycle Life. Figure 4 shows the cycle lives of electrodes CP3 and CC. The results indicate that the cycle lives of the electrodes are related to the kind of technique used: the longest cycle life was achieved for the electrode built with a train of pulses. For example, after 300 charge-discharge cycles (100% depth of discharge, DOD), the electrode CP3 still kept 96% of the initial capacity, which was much higher than that of any other electrode.

Each cycle begins with 150 min charge at a C/2 rate followed by a C/2 rate discharge to 0.2 V versus Hg/HgOss.

A stable discharge capacity is observed with cycle life until cycle 500, after which the capacity drops sharply. One of the reasons is electrode swelling, which changes the electrode thickness after several numbers of cycles. Therefore, the cycle life of the electrode can be correlated with the electrode expansion: the greater the electrode swelling is, the faster the capacity decay of the electrode results.

3.5. Physical Properties of Nickel Hydroxide. Zhang and Hou-Tian have performed the impregnation by applying intermittent currents, ON:OFF pulses, with the aim of reducing the activation and polarization by concentrating polarization on the cathode. In this way, smaller active material particles were formed, thus increasing the active surface area of the electrode and achieving greater use of the deposited material [31].

Photomicrographs (Figures 5(a) and 5(b)) show, with different magnifications, the morphology of the porous substrate prior to impregnation.
The morphology of the deposited material is shown in Figures 6 and 7.

The deposits made by applying a continuous current Figure 6(a) cover the total nickel mesh and are more compact than those deposited through a train of pulses (Figure 7(a)).

Figure 6 shows that the material is compact and its composition corresponds to the active material that contains Ni and Co (see Table 2).

The loading (g cm$^{-3}$) from sample CC (continuous current) was 10% larger than that of sample CP3 (pulse current). The surface morphology of the deposited material in the first sample is homogeneous and compact. Although we achieve a greater loading, this value does not guarantee a better nickel electrode capacity.

In the case of pulse-current deposition, the active material is deposited along the shape of the porous substrate and leaves visible gaps or pores. This leads to an increased deposition of the surface active material, achieving a higher utilization rate. It can also be observed that the deposited material has a spongy appearance, generating an electrode with a larger active surface area and smaller particle size.

Table 2 presents the EDS results of both nickel electrodes with cobalt added. The cobalt present in the active material of the electrode was in the same proportion as in the impregnation solution.

<table>
<thead>
<tr>
<th>Table 2: Chemical properties of different types of electrode samples.</th>
</tr>
</thead>
<tbody>
<tr>
<td>Analysis (% w/w)</td>
</tr>
<tr>
<td>Ni</td>
</tr>
<tr>
<td>----------------</td>
</tr>
<tr>
<td>Electrode CC</td>
</tr>
<tr>
<td>Electrode CP3</td>
</tr>
</tbody>
</table>

4. Conclusions

All electrodes that were electrochemically impregnated with pulse current showed optimum capacity, similar to loading values found in the literature, with the active material producing a larger active surface area. When the impregnation current is continuous, the active material forms a homogeneous and compact surface with optimum loading but poor performance during discharge. Furthermore, the electrodes that were impregnated with different pulse currents produce loading similar to that of the electrodes impregnated with continuous current, although their active material is deposited following the shape of the porous substrate. This type of deposition produces a larger active surface area that has the highest utilization percentage. The particle size influences the electrochemical performance. When the particle size decreases, the amount of adsorbed water molecules on the surface of the particles increases. This
surface water is thought to improve nickel hydroxide particle wettability, thus resulting in an enhanced proton transport within the active mass during the charge-discharge process and a better utilization of the electrode material.

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References


Figure 7: SEM micrographs of the electrode surface (CP3), with nickel hydroxide as the active material; (a) 120x and (b) 3500x.


